Optical absorption of nickel (II) in Na₂O-NaCI-B₂O₃ and Na₂O-NaBr-B₂O₃ glasses

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The optical absorption spectra of nickel(II) have been studied in $Na_2O-NaX-B_2O_3$ glasses X = CI or Br); the absorption bands have been interpreted in terms of Ligand Field Theory. In halide-free B_2O_3 -rich glasses nickel(II) is octahedral, and halide substitution is accompanied by the formation of a tetrahedral nickel(II)-oxide-halide complex. In alkali-rich glasses nickel(II) is square planar and/or tetrahedral and no halide substitution can be observed. It is suggested that the halide ions are unable to substitute for oxygen in the predominantly covalent nickel(II)-oxygen bonds of the complex formed in basic glasses.

1. Introduction

Use of nickel as a colourant in glass is common; sometimes a small concentration of nickel is also used as decolourizer in glass industry. The colour of nickel containing glasses is known to be critically dependent on the composition and the thermal history of the glass. In this paper we present experimental evidence that the colour of nickel containing glass can also be altered by making small additives with the batch.

Nickel usually occurs in glass as nickel(II) in octahedral, square planar and tetrahedral symmetry depending upon the composition of the glass [1]. High-spin octahedral nickel(II) which occurs in low-alkali borate glasses is olive green; square planar nickel(II) is usually greyish brown and occurs in Na2O-SiO2 and high-alkali Na₂O-B₂O₃ glasses, whereas in high-alkali K_2O -SiO₂ and K_2O -B₂O₃ glasses nickel(II) is tetrahedral and imparts a pink colour. Numerous investigations on mixed ligand nickel(II) complexes have been reported in molten salt and in co-ordination compounds. When alkali halides or alkali sulphides are added to a glass containing a transition metal ion, anionic substitution around the transition metal ion takes place with a subsequent change in symmetry of the complex [2]. In this paper we report the chloride and bromide substitution around nickel(II) and the subsequent changes in symmetry of the complex in $Na_2O-B_2O_3$ glasses.

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2. Experimental

All the raw materials (Na₂CO₃, NaCl, NaBr and H₃BO₃) were of "analar" quality; 0.50 wt % nickel was added as green NiO powder to the batch. Ten grammes of glass were melted at a time in a Pt + 2% Rh crucible at 1000°C in an electric furnace for four hours. Halide containing glasses were melted by partially substituting Na₂CO₃ by NaCl or NaBr in the batch; these glasses were stirred with a thick platinum wire for uniform distribution of halide throughout the melt.

All the constituents of the glass were estimated chemically. Nickel was estimated colorimetrically with dimethyl glyoxime [1]. Absorption spectra of 0.5 to 0.05 cm thick polished samples were measured on a Cary 14 spectrophotometer. In every case a blank glass of corresponding composition and thickness but without nickel was prepared and its absorption spectrum measured. In all the reported spectra of nickel(II) the blank absorption was subtracted.

3. Results

The composition of glasses studied in this investigation is shown in Table I. Absorption spectra of nickel(II) in Na₂O-NaCl-B₂O₃ glasses are shown in Fig. 1a and b. Nickel (II) in 16Na₂O,84B₂O₃ glass is octahedral with three main absorption bands at 7200, 13 200 and 24 100 cm⁻¹. On small (up to 0.17 mol NaCl per litre tia

Glass no.	Mol % B ₂ O ₃	Density (g cm ⁻³)	Moles per litre glass			
			Total Ni	NaCl	NaBr	
1	84	1.7631	0.151	0.000	·	
2	84	1.7620	0.143	0.126		
3	84	1.7621	0.139	0.164		
4	84	1.7600	0.147	0.169		
5	84	1.7521	0.148	0.881		
6	84	1.7001	0.138	2.486	_	
44	92	1.9838	0.148		0.000	
45	92	1.9854	0.137	_	0.042	
46	92	1.9879	0.137		0.060	
47	92	1.9878	0.137	_	0.073	
48	92	1.9853	0.139		0.086	
49	92	1.9869	0.147	_	0.179	
50	92	1.9888	0.147	_	0.224	
51	92	1.9887	0.143		0.231	
52	84	2.0801	0.143	_	0.000	
54	84	2.0754	0.143	_	0.092	
56	84	2.0673	0.147	_	0.125	
58	84	2.0711	0.148		0.204	
76	75	2.2300	0.129	_	0.000	
78	75	2.2360	0.117	—	0.769	
83	75	2.2287	0.147		1.874	

 TABLE I Composition of glasses (after chemical analysis)



Figure 1 (a) and (b) Optical absorption spectra of nickel(II) in $(16 - x)Na_2O$, 2xNaCl, $84B_2O_3$ glasses.

glass) substitution of NaCl for Na₂O in the glass, a shoulder is developed on the low energy side which may be represented as a separate set of bands buried in the more intense neigh-1058 bouring band; the other bands at 13 200 and 7200 cm⁻¹ do not show any significant change. With larger amounts of NaCl in the glass (Fig. 1b) an altogether new set of bands are developed



Figure 2 Optical absorption spectra of nickel(II) in $(8 - x)Na_2O$, 2xNaBr, $92B_2O_3$ glasses.



Figure 3 Optical absorption spectra of nickel(II) in (16 - x)Na₂O, 2xNaBr, 84B₂O₃ glasses.

around 7900, 15 800 and 21 000 cm^{-1} characteristic of square planar and tetrahedral nickel(II) complexes.

Absorption spectra of nickel(II) in different Na₂O-NaBr-B₂O₃ glasses are shown in Figs. 2 to 4. In the glass $8Na_2O.92B_2O_3$ (Fig. 2) a very small amount of bromide (≥ 0.08 mol NaBr per litre glass) produces the absorption bands around 7000 and 14 000 cm⁻¹ characteristic of tetrahedral nickel(II); the intensity of all the tetrahedral bands increase with increasing amounts of bromide (≥ 0.8 mol NaBr per litre glass) in these glasses, a strong tail of charge transfer bands due to nickel(II) tetrahedral



Figure 4 Optical absorption spectra of nickel(II) in (25 - x)Na₂O, 2xNaBr, 75B₂O₃ glasses.

complex intrudes into the visible region increasing the blue end absorption of the glass. The effect of bromide substitution in $16Na_2O_84B_2O_3$ glass containing nickel(II) is very similar; only a larger amount of bromide is necessary to produce the tetrahedral bands. In $25Na_2O$, $75B_2O_3$ glass no tetrahedral nickel(II) is formed even with 1.87 mol bromide/litre glass (Glass 83). Fig. 5 shows the absorption bands of tetrahedral nickel(II) in different glasses with oxide and oxide-halide coordination; and Fig. 6a and b show the absorption spectra of crystallines NiCl₂ and NiBr₂ respectively.

4. Discussion

Nickel(II) is $3d^8$; the relevant part of the energy level diagram for a d^8 system is shown in Fig. 7a and b. Both in octahedral and in tetrahedral



Figure 5 Optical absorption spectra of tetrahedral nickel(II) in different glasses with oxide and oxide-halide co-ordination.



Figure 6(a) Optical absorption spectra of crystalline NiCl₂. (b) Optical absorption spectra of NiBr₂ (crystalline).

symmetry three spin-allowed transitions are expected with the following energies [3]:

Octahedral

$${}^{3}A_{2}(F) \rightarrow {}^{3}T_{1}(F) = \frac{15}{2} B + 15Dq - \frac{1}{2} (225B^{2} + 180BDq + 100Dq^{2})^{\frac{1}{2}}$$
$$\rightarrow {}^{3}T_{1}(P) = \frac{15}{2} B + 15Dq + \frac{1}{2} (225B^{2} + 180BDq + 100Dq^{2})^{\frac{1}{2}}$$
$$\rightarrow {}^{3}T_{2}(F) = 10Dq$$

Tetrahedral

$${}^{8}T_{1}(F) \rightarrow {}^{3}T_{1}(P) = (225B^{2} + 180BDq + 100Dq^{2})^{\frac{1}{2}}$$

$$\rightarrow {}^{3}A_{2}(F) = \frac{15}{2}B + 15Dq + \frac{1}{2}(225B^{2} + 180BDq + 100Dq^{2})^{\frac{1}{2}}$$

$$\rightarrow {}^{3}T_{2}(F) = \frac{15}{2}B + 5Dq + \frac{1}{2}(225B^{2} + 180BDq + 100Dq^{2})^{\frac{1}{2}}$$

Glasses 1, 44 and 52 are green; from Figs. 1a, 2 and 3 it can be seen that three absorption bands are obtained around 7000, 13 000 and 24 000 cm⁻¹ and these are assigned to ${}^{3}A_{2}(F) \rightarrow {}^{3}T_{1}(F)$, ${}^{3}A_{2}(F) \rightarrow {}^{3}T_{1}(F)$, and ${}^{3}A_{2}(F) \rightarrow {}^{3}T_{1}(P)$ transitions respectively. With these assignments the appropriate algebraic equations have been solved to get ligand field parameters (Δ_{0} and B) **1060**



Figure 7 (a) Energy level diagram of d^8 system in octahedral field. (b) Energy level diagram of d^8 system in tetrahedral field.

in different glasses and crystals, and these are given in Table II along with those for $[Ni(H_2O)_6]^{2+}$ and $[Ni(en)_3]^{2+}$ complexes in aqueous solution. Δ_{oct} in 25Na₂O,75B₂O₃ glass apparently seems to be smaller than that in 8Na₂O,92B₂O₃ and 16Na₂O,84B₂O₃ glasses. It may be that the symmetry of nickel(II) complex in 25Na₂O, 75B₂O₃ glass is D_{4h} rather than C_{3v}; this will be discussed in a later section.

The middle absorption band of octahedral nickel(II) around 13 000 cm⁻¹ corresponding formally to the transition: ${}^{3}A_{2g}(F) \rightarrow {}^{3}T_{1g}(F)$ is double peaked. According to Jøorgensen [4] this is due to the closeness of a ¹E state coming from ¹D; spin-orbit coupling mixes the ${}^{3}T_{1}(F)$ and ¹E states and splits the band. Ballhausen *et al* [5] have calculated the complete energy level diagram for 3d⁸ system with spin-orbit coupling included and shown that the ¹E level never does approach "very closely" to any triplet level

Complex	Optical transitions (cm ⁻¹)			$\Delta_{\rm oct}$ (cm ⁻¹)	B (cm ⁻¹)
	${}^3\mathrm{A}_2 \to {}^3\mathrm{T}_2$	${}^{3}A_{2} \rightarrow {}^{3}T_{1}(F)$	$^{3}A_{2} \rightarrow ^{3}T_{1}(P)$	_	
$[Ni(H_2O)_6]^{2+}$	8 500	14 000	25 000	8 500	940
$[Ni(en)_3]^{2+}$	11 500	18 500	30 000	11 500	850
Nickel(II) in $8Na_2O$, $92B_2O_3$ glass	7 200	13 200	24 100	7 200	1 047
Nickel(II) in 16Na ₂ O, 84B ₂ O ₃ glass	7 700	14 000	25 000	7 700	1 060
Nickel(II) in $25Na_2O$, $75B_2O_3$ glass	6 500	12 900	23 800	6 500	1 146
NiCl ₂	7 720	12 400	22 065	7 720	750
NiBr ₂	7 420	12 820	20 290	7 420	650

TABLE II Optical transitions and ligand field parameters of some octahedral nickel(II) complexes

However, it must be pointed out that the calculations of Ballhausen are strongly dependent upon the choice of the values of Racah parameters, B and C, and the spin-orbit coupling parameter, λ .

 $8Na_2O,92B_2O_3$ and 16Na₂O,84B₂O₃ In glasses containing halides, in addition to octahedral bands two sets of new major bands are developed around 8000 and $15\,000$ cm⁻¹; the intensity of these two band systems increases with increasing halide content of the glass. These two bands were assigned to ${}^{3}T_{1}(F) \rightarrow {}^{3}A_{2}$ and ${}^{3}T_{1}(F) \rightarrow {}^{3}T_{1}(P)$ transitions respectively. With these assignments the appropriate algebraic equations have been solved for Δ_{tet} and B, and these are shown in Table III. Δ_{tet} which turns out to be around 4000 cm⁻¹ is about $4/9 \Delta_{oct}$ in these glasses; unfortunately this region of spectrum is masked by strong hydroxyl absorption in these glasses (due to residual water in the glass from batch materials) and thus this transition could not be experimentally identified. It is clear from Table III that Δ_{tet} is larger and B is smaller in

bromide containing glasses than those in pure oxide or oxide-chloride containing glasses. This decrease in B indicates that nickel(II)—bromide bonds are more covalent that the corresponding nickel(II)—chloride or nickel(II)—oxide bonds in glass. The same trend has also been observed in the case of octahedral nickel(II) complexes in glass (see Table II).

Comparing absorption spectra of nickel(II) in halide containing glasses with those of crystalline pure NiCl₂ and NiBr₂ it is evident that while pure NiCl₂ and NiBr₂ are distorted octahedra [6], the nickel(II) complex in the halide containing glasses occur with a mixture of oxide and halide ligands and is in tetrahedral symmetry. It may be seen from Fig. 5 the tetrahedral nickel(II) absorption bands in glasses containing bromide are shifted to lower energies in comparison to oxide and oxide-chloride complexes. This is consistent with the spectrochemical series where oxide>chloride>bromide.

Unlike B_2O_3 -rich glasses no striking change in the nickel (II) absorption spectra occurred in

Complex Optical transitions (cm⁻¹) Δ_{tet} (cm⁻¹) B (cm⁻¹) ${}^{3}T_{1} \rightarrow {}^{3}T_{2}$ ${}^{3}T_{1} \rightarrow {}^{3}A_{2}$ ${}^{3}T_{1} \rightarrow {}^{3}T_{1}(P)$ Ni(Ph₃AsO)₂Br₂ (3435)* 7500 15 000 4065 813 Nickel(II) in 16Na₂O, 84B₂O₃ (3615)* 7900 15 800 4285 857 glass containing chloride Nickel(II) in 8Na₂O,92B₂O₃ (3250)* 7100 14 200 3850 770 glass containing bromide Nickel(II) in 16Na₂O,84B₂O₃ (3740)* 7700 14 600 3960 792 glass containing bromide

TABLE III Optical transitions and ligand field parameters of some tetrahedral nickel(II) complexes

*Calculated values

25Na₂O,75B₂O₃ glasses even with much larger amounts of bromide. This may arise due to two reasons: (a) nickel(II) in this glass is four coordinated and covalently bonded [1], and thus bromide substitution is not favourable. It is known with other transition metal ions like copper(II) [10], vanadium(IV) [11] in Na₂O- B_2O_3 glasses that the metal-oxygen covalency increases sharply above 20 mol % Na₂O in the glass; (b) the activity coefficient of bromide in this glass may be too small to produce noticeable substitution with the limited amount of bromide added in these glasses (2d). Although no activity measurement of NaBr in Na₂O-B₂O₃ melt has yet been reported, the solubility of NaCl in $Na_2O-B_2O_3$ melts is known to increase sharply with Na₂O content of the melt [12].

Absorption due to square planar nickel(II)

Nickel(II) having d⁸ electronic configuration is prone to square planar complex formation with simple ligands [7]. The five degenerate d-orbitals of the uncomplexed metal ion split into four different levels in a square planar complex. The general molecular orbital energy levels for square planar complexes in which the ligands have no π -orbital system (chloride, bromide, oxide, etc.) is shown in Fig. 8. Almost all square planar complexes of nickel(II) with simple ligands are diamagnetic and the ground state is ${}^{1}A_{1g}$ Three spin-allowed d-d type transitions, corresponding to b_{2g} (π^*) $\rightarrow b_{1g}$ (δ^*) ($^1A_{1g} \rightarrow {}^1A_{2g}$), $a_{1g} (\delta^*) \rightarrow b_{1g} (\delta^*) ({}^{1}A_{1g} \rightarrow {}^{1}B_{1g}) \text{ and } e_g (\pi^*) \rightarrow$ $b_{1g} (\delta^*) ({}^{1}A_{1g} \rightarrow 1E_g)$ are expected. All these are parity forbidden as electric dipole transitions. In addition two allowed charge transfer transitions may be anticipated corresponding to:

$$\begin{array}{c} b_{2u}\left(\pi\right) \rightarrow b_{1g}\left(\delta^{*}\right)\left({}^{1}A_{1g} \rightarrow {}^{1}A_{2u}\right) \\ \text{and } e_{u}\left(\pi^{b}\right) \rightarrow {}^{b}1g\left(\delta^{*}\right)\left({}^{1}A_{1g} \rightarrow {}^{1}E_{u}\right). \end{array}$$

These are transitions from molecular orbitals essentially localized on the ligands to molecular orbitals essentially localized on the metal atom.

All these transitions in square planar nickel(II) complexes have been observed [8]. One distinctive feature of the square planar nickel(II) complex when compared to the octahedral complex is the development of a strong shoulder around 20 000 cm⁻¹ [1, 9]. It is clear from Figs. 2 and 3 that with smaller amounts of halides in $8Na_2O_32B_2O_3$ and $16Na_2O_84B_2O_3$ glasses, prior to tetrahedral nickel(II) formation, the shoulder around 20 000 cm⁻¹ gets stronger with



Figure 8 Qualitative molecular orbital energy level scheme for square planar d^8 metal complexes with no intra-ligand π -orbital system (after Gray and Ballhausen).

halide content indicating square planar nickel(II) formation in these glasses.

Appendix

The symbols used in this paper are all standard spectroscopic symbols. For those not conversant in this field the following brief explanations are given:

A, B and C: Racah parameters

10Dq: ligand field strength

Energy of any "state" of an ion in an electrostatic field of strength 10Dq, can be expressed in terms of A, B, C and 10Dq (as written in the equations on page 1060) by solving the appropriate Tanabe-Sugano matrices.

- λ : Spin-orbit coupling; this originates from coupling of spin angular momentum of electron with orbital angular momentum.
- δ and π : These indicate bonding molecular orbitals formed by combination of suitable atomic orbitals.
- δ^* and π^* : These are antibonding orbitals. δ and π indicate particular type of overlap of electronic wavefunctions. Antibonding orbitals of any kind are at higher energy (unstable) than the corresponding bonding orbitals.

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